

Highlights

Synthetic generation of vibroacoustic modulation signals for structural health monitoring

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- Generation of VAM from static ultrasonic measurements for a robust and independent SHM.
- Decoupling VAM from actual ambient vibration makes measurements comparable over time.
- Data reduction of 99.99% compared to traditional VAM evaluation.
- Feasible with undersampled signals resulting in reduced hardware requirements.
- Enabling the application of VAM for actual structures.

Synthetic generation of vibroacoustic modulation signals for structural health monitoring

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Abstract

The vibroacoustic modulation (VAM) is a nonlinear ultrasonic testing method that utilizes the modulation of a high-frequency/low-amplitude probe wave with a low-frequency/high-amplitude pumping vibration, resulting in high sensitivities to damages in the structure. However, applying the method outside of the laboratory as a structural health monitoring system for actual structures is challenging, since the amplitude and frequency of the ambient vibrations—which is ideally utilized as pumping vibration—fluctuates over time. To circumvent this, we present a synthetic generation of the VAM signal using only the probe measurements, acquired at two (or more) stress levels of the structure when a steady state is reached. We could show that only 16 values (8 measured values of only two stress levels with a sampling

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frequency of 1/20 of the Nyquist frequency) are required to generate the often calculated Modulation Index with a mean deviation of 0.97 % to the expected dynamic measured values for glass fiber reinforced composites and 1.86 % for the aluminum specimens, which is negligible compared to a typical increase of the Modulation Index of 10–30 dB in case of severe damage. Even undersampled measurements at each stress level can be used without sacrificing accuracy, which reduces the sensing requirements for the sensor nodes. Moreover, this method decouples VAM from the actual need for a constant recurrent frequency and amplitude of the natural vibration in order to reliably compare measurements throughout the lifetime. Hence, this work aims to open the possibility of ultimately applying VAM to assess the structural health of complex structures.

Keywords: Damage Detection, Vibroacoustic Modulation, Structural Health Monitoring, Nondestructive Testing, Nonlinear Acoustics, Lamb Waves

1. Introduction

The evaluation of structural integrity is a concern for all engineering applications. Hence, Non-Destructive Testing (NDT) is crucial for reliable operations. Within the multitude of NDT methods, especially nonlinear ultrasonic methods attain high sensitivities for small defects, which often remain concealed in linear ultrasonic methods. [1] A highly sensitive nonlinear technique is the Vibroacoustic Modulation (VAM) analysis, which has been the subject of first studies in the 1990s [2, 3, 4, 5, 6, 7], recent publications [8, 9, 10, 11, 12, 13, 14, 15, 16, 17, 18, 19] and many more in between. The

10 method utilizes the modulation of a high-frequency ultrasonic Lamb wave as
11 a carrier (or probe) signal $s_c(t)$ with a high-amplitude pump vibration $s_p(t)$
12 for which an already present ambient signal or an artificially generated wave
13 can be utilized. VAM has been applied to many applications, from moni-
14 toring fatigue crack growth in aluminum specimens,[20, 21, 22, 23] detection
15 of impacts and delaminations in composite structures [24, 25, 26] or identi-
16 fication of adhesive contaminations. [26, 27, 28] A comprehensive review of
17 already tested materials and promising applications is given by Pieczonka
18 *et al.*[29] and in the book chapters [30, 31]. Unfortunately, most research on
19 VAM has been performed exclusively on laboratory-scale specimens where
20 both vibrations (pump and carrier) are artificially introduced. Especially the
21 pump wave, which is enabling the high sensitivity of VAM, is particularly
22 identified and optimized [32], or even the full data of a frequency sweep over
23 a range is used.[33] Hence, the utilization of VAM in many works could be
24 classified as an NDT method rather than the Structural Health Monitoring
25 (SHM) method it could become.

26 To monitor and estimate the structural health of large civil or engineer-
27 ing structures, the excitation of a defined pump vibration is cumbersome
28 and requires powerful actuators. However, these actuators depend on strong
29 external energy sources and therefore prevent easy integration of the VAM
30 into the structure. Therefore, it would be beneficial to use already existing en-
31 vironmental vibrations. For optimal damage detection and the possibility of
32 damage localization [19], data has to be acquired at several locations with a
33 network of multiple sensors. Therefore, the optimal placement of sensors,[34]
34 and a wireless application of energy harvesting sensor nodes for data mea-

35 surement and transmission are relevant and actively developed. [35, 36, 37]
36 However, the required energy consumption for the high-frequency excitation,
37 high sampling rate measurements, complex computational evaluation, and
38 the transmission of results pose severe difficulties for VAM applications on
39 large structures. Using the short-time Fourier-transform (STFT) on a highly
40 undersampled signal has shown great potential to reduce the energy for the
41 measurement and data evaluation. [38] Nevertheless, the VAM method—as
42 applied up to now—requires the excitation of an ultrasonic signal s_c and the
43 ambient vibration s_p to be constant during the measurement and more im-
44 portantly for the following measurements as well. However, such ideal pump
45 vibrations are not evident in actual civil or engineering structures. Vibra-
46 tional data acquired at the Köhlbrandbrücke in Hamburg (Germany) shows
47 that the ambient vibration is a superposition of frequencies in the range 10–
48 20 Hz. [38, 39] Additionally, these ambient structural vibrations depend on
49 many environmental influences, e.g., winds, temperature, traffic, and sea-
50 sonal aspects. Hence, the eigenmodes and resulting dominant frequencies in
51 the structure fluctuates, which influences the pump vibration and hampers
52 the applicability of VAM.

53 To overcome these limitations, we present a new method to acquire the
54 data for the VAM method. The novelty of this work is the artificial recon-
55 struction of a VAM signal from several measurements of only the carrier
56 vibration at different stress states of the structure, instead of analyzing the
57 carrier and pump vibrations simultaneously. The method is based on the
58 assumption that the carrier frequency is much higher compared to the pump
59 frequency and hence, requires only a few periods T_c of the ultrasonic carrier

60 s_c at these stress levels. These carrier measurements are combined into a
61 synthetic signal, from which the damage state can be assessed, for example,
62 with the conventional Modulation Index (MI) [40] or other evaluation meth-
63 ods. [29, 38] The presented method is the first which actively decouples the
64 VAM analysis from the amplitude and frequency of the pump wave. Hence,
65 the damage state could be assessed, even if the ambient conditions change.
66 The synthetic VAM increases the comparability for a reliable and precise
67 health monitoring. Ultimately, the presented procedure leads to a signifi-
68 cant data reduction of each measurement, thus reducing the requirements on
69 the sensor nodes possibly enabling a more sophisticated multi-sensor VAM
70 approach in the future which is even able to locate damages.

71 **2. Methodology**

72 *2.1. Vibroacoustic modulation*

73 The VAM analysis is an active acoustic method that utilizes the non-
74 linear response of the material to detect damages and can be applied to
75 NDT and potentially SHM. Typically, the specimen or structure is excited
76 simultaneously with a low-frequency vibration s_p (often referred to as pump
77 wave) and an ultrasonic vibration s_c (probe or carrier wave). The ultrasonic
78 carrier wave with a frequency of f_c is introduced by piezoceramic actuators
79 attached to the specimen. However, methods for introducing a pump wave
80 s_p with frequency f_p differ in the literature and are either orthogonal or lon-
81 gitudinal excitations. Electromagnetically shakers [23, 41] or piezoceramics
82 [22, 26, 42] typically introduce the wave orthogonal, while servohydraulic
83 testing machines [21, 43, 44, 45] introduce it longitudinal.

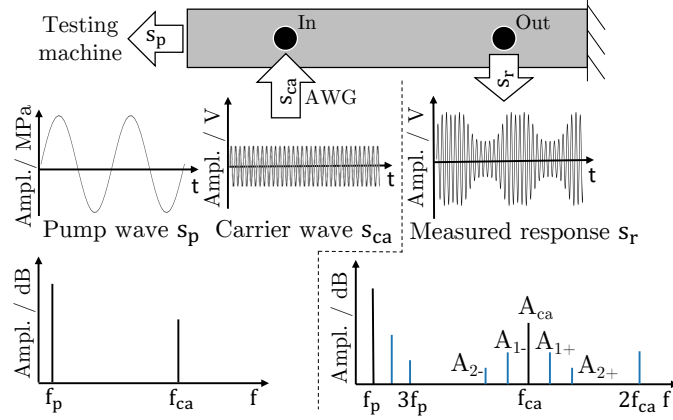


Figure 1: Schematic experiment setup. The pumping wave s_p is introduced by a servo-hydraulic pulsing machine, while the high frequency carrier s_c is generated by a piezoceramic actuators, depicted as black disks and driven by an arbitrary waveform generator (AWG). (adapted from [27])

84 The pump wave periodically alters the stress state in the specimen. This
85 variation is particularly enhanced around inhomogeneities or defects and pe-
86 riodically affects the propagation of the ultrasonic waves. The system re-
87 sponse signal s_r is typically measured by a second piezoceramic or a laser
88 doppler vibrometer. [24, 44, 46] The general approach of a VAM analysis is
89 schematically illustrated in Figure 1. The initiation and growth of a damage
90 alters the stress distribution, which further affects the nonlinear response of
91 the specimen. Consequently, the higher harmonics ($k f_p \wedge k f_c \mid k \in \mathbb{N}$) and
92 sidebands around the carrier wave ($f_c \pm k f_p \mid k \in \mathbb{N}$) will increase, the latter
93 representing a stronger modulation. [29] These components can be seen in
94 the frequency domain of the measured signal, calculated by a Fast Fourier
95 Transform (FFT). The appearance of sidebands is associated to a clapping
96 mechanism of the defects and changes in the stress distribution around the

97 tips of the cracks. Ref. 29 associated the highest longitudinal stress of the
98 pump wave with the lowest signal strength in the system response. However,
99 the numerical and experimental results from Dorendorf *et al.*[44] suggest that
100 this assumption is dependent on the excited carrier frequency with regard to
101 the eigenmodes of the specimen. A thorough mathematical explanation of
102 the occurrence of sidebands is found elsewhere. [22, 44, 26, 47, 48, 49]

103 VAM is commonly evaluated by calculating the ratio of the sideband
104 amplitudes to the carrier amplitude, resulting in the Modulation Index (MI)
105 and given in decibels:[40, 43, 50]

$$\text{MI} = 20 \cdot \log_{10} \left(\frac{A_{-1} + A_{+1}}{2A_c} \right) \quad (1)$$

106 Here, A_c is the spectral amplitude of the carrier wave s_c . The sideband
107 amplitudes A_k are differentiated by the numeric index k that indicates its
108 corresponding frequency. A summation or averaging of several sidebands can
109 be performed to extract more information from each measurement. Addition-
110 ally, averaging over multiple frequencies is commonly suggested to eliminate
111 variations in MI due to resonances and antiresonances of the tested speci-
112 men. [7, 27, 43, 45]

113 Furthermore, the modulation in a VAM measurement of any structure
114 is a superposition of amplitude modulation and angular modulation (phase
115 and frequency) of the frequencies present in the structure and partly caused
116 by a periodic change of wave propagation characteristics due to nonlinear-
117 ities in wave propagation and the signal path length. Hence, separating the
118 measured signal into different modulation types has shown benefits, as they
119 could be attributed to different types of damage in a sample. [20, 50, 43] Hu
120 *et al.* 50 proposed the separation by the *Hilbert-Huang transformation* and

121 suggests that the modulated signal amplitude has a higher correlation with
122 the size of the crack. In contrast, Refs. 20 and 43 concluded that apply-
123 ing the *In-phase Quadrature Homodyne Separation algorithm*, the frequency
124 modulation indicates fatigue damage for aluminum specimens at an earlier
125 point. The analytical model of Dorendorf *et al.*[44] indicates the influence
126 of each nonlinearity (e.g. damages, contaminations, among others) on both
127 types of modulation.

128 Finally, for inspections of large structures, it is beneficial to utilize the
129 ambient vibrations which are already present in the structure and originate
130 from environmental influences (winds, water flow, etc.) or other load fac-
131 tors (traffic, rotor movement, engines, pumps, etc.). [7] The application of
132 these vibrations allows low energy measurements, but also results in mul-
133 tiple unknowns regarding the low-frequency pump vibrations that do not
134 have a constant amplitude and frequency over time. Therefore, it hampers
135 the application of the separation algorithms mentioned above and makes
136 the VAM cumbersome to reproduce for standardized measurements on these
137 large structures.

138 2.2. Synthetic signal generation

139 Traditionally, a sinusoidal pump vibration is used to alter the stress in
140 the structure, which widens the cracks periodically and results in signal mod-
141 ulation. It was shown by Dorendorf *et al.*[44], that the measured amplitude
142 modulation of VAM oscillates harmonically between measurements of only
143 the excited carrier at the highest and lowest stress of the pump vibration.
144 For most frequencies, the extreme stresses coincide with either the minimal
145 or maximal measured voltage on the piezoceramics. The amplitude hereby

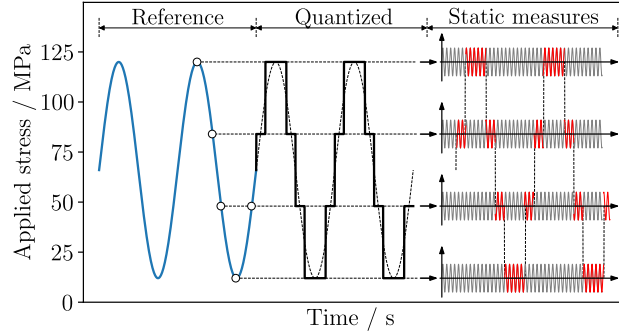


Figure 2: The conventional sinusoidal low-frequency excitation (blue) can be quantized into discrete stress levels (black). The synthetic VAM is composed of measurements from only the high-frequency excitation at each of the stress levels (right). The red areas mark the chunks of each measurement, which are concatenated into the composed synthetic signal.

146 depends on the evaluated frequency with respect to the eigenmodes of the
 147 corresponding specimen.

148 Since $f_c \gg f_p$, it can be assumed that the amplitude of the measured
 149 modulated signal at each point in time can be approximated by the steady-
 150 state response of the carrier wave s_c at the corresponding stress state. There-
 151 fore, the main concept behind the synthetic VAM is to quantize the pump
 152 vibration into a distinct number of discrete stress levels n —similar to analog-
 153 to-digital signal processing—and acquire only the high-frequency response at
 154 these individual levels σ_l , where l indication actual stress at this level.

155 This procedure is visualized in Figure 2 for $n = 4$ individual stress levels.
 156 The pump vibration—which is traditionally used for VAM—is shown on the
 157 left side. The dots on the pump wave indicate the mapped sampling points
 158 for the quantization of the pump vibration. In this example, equidistant
 159 stress steps were chosen, but even random points can be feasible as well.

160 The quantized pump wave is shown in Figure 2 (mid) and at each step of
161 this quantization, only the carrier response is measured (right).

162 The synthetic signal is composed of concatenated signal fragments, ex-
163 tracted from the carrier measurements as defined by the quantized pump
164 vibration (Figure 2 right). Due to the concatenation, this method results in
165 non-continuous transitions between each segment and a resulting quantiza-
166 tion error. Intuitively, the quantization error increases with fewer levels n
167 and is explained in more detail later. Furthermore, this method will have
168 a small error due to the assumption that a linear relationship between the
169 strain of the specimen and the amplitude of the vibration is evident, resulting
170 in a pure sinusoidal envelope of the resulting signal. The error in MI, which
171 is the deviation from the expected offset and results of this assumption, will
172 be discussed based on the results of experimentally tested specimens and the
173 results from numerical finite element simulations in the following sections.
174 In Figure 3 (left) the synthetic generation of an amplitude-modulated sig-
175 nal is shown for the two representative numbers of stress levels $n = 2$ and
176 $n = 3$. The pump wave s_p of the amplitude modulation (Equ. A.3 in the
177 Appendix) was quantized to n stress levels at which the carrier signal was
178 computed and the points were extracted to calculate the static cases. Note
179 that the envelope of the modulated signals is almost symmetrical, hence it is
180 irrelevant whether the upper or lower half is shown in this figure.

181 Amplitude modulation is visible for all synthetic signals and follows the
182 dynamic reference, illustrated as a black envelope. By increasing the num-
183 ber of steps n , the synthetic signal converges towards the dynamic system
184 response. However, the utilization of more steps also comes with the price

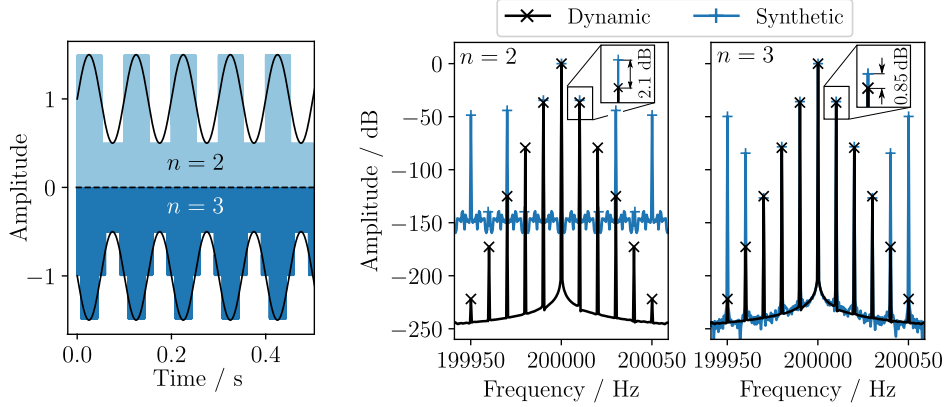


Figure 3: Comparison of an amplitude and phase modulated signal to the synthetic generated signals from $n = 2$ and 3 static levels. The time domain (left) and also the frequency domain (right) show a higher precision of the approximation with an increased n .

185 of more required measurements and data points, which has to be stored and
 186 transmitted. Hence, more relevant for the application of sensor networks
 187 is the reduction of steps to the minimum of ($n = 2$), which results in the
 188 composition of a square wave (Figure 3, upper left).

189 The Fourier Transform \mathcal{F} of a composed synthetic signal from only two
 190 steps—resulting in a square wave (sq)—shows unique characteristics in the
 191 frequency domain (eq. 2) as known from signal processing. [51]

$$\mathcal{F} \{ \text{sq}(2\pi f_p) \} = \begin{cases} \frac{4}{l\pi}, & \text{for } |f| = lf_p, \text{ with } l = 1, 3, 5, \dots \\ 0, & \text{otherwise} \end{cases} \quad (2)$$

192 Due to the particular nature of square waves, only odd sidebands ($l =$
 193 $1, 3, 5, \dots$) appear in the frequency domain, where the first sideband is in-
 194 creased by a factor of $4/\pi$, compared to the dynamic signal response from a
 195 sinusoidal pumping wave s_p . Furthermore, the following odd sidebands have
 196 the amplitude of $4/(l\pi)$.

197 If we apply a square wave $s_p = \text{sq}(2\pi f_p)$ for the signal modulation, the
 198 relationship in eq. 2 also applies to the sidebands around the carrier. Hence,
 199 the intensity of the carrier is equal for a square modulation and sinusoidal
 200 modulation. Consequently, the MI of a synthetic signal from $n = 2$ is offset
 201 by $20 \cdot \log_{10}(4/\pi)$, which equals 2.098 dB as shown in the inset of Fig. 3
 202 (right). Although the amplitude of the carrier matches the dynamic signal in
 203 both cases, an offset can be seen for the first sidebands of the square signal
 204 of the value mentioned above of $4/\pi$. The amplitude of further sidebands
 205 decreases linearly (cf. eq. 2). For $n = 3$, the synthetic signal closely resembles
 206 the dynamic measurement, resulting in a high similarity of the first three
 207 sidebands in the frequency domain.

208 Most intuitively, VAM depends on the sampling rate f_s , which directly
 209 influences the maximal measurable frequency. Nyquists theorem states that
 210 the sampling rate should minimally be $f_s > 2 \cdot f_{\max}$. For VAM measure-
 211 ments, f_{\max} represents the frequency of the highest sideband that should be
 212 evaluated. There are already some techniques (e.g., undersampling of the
 213 measured signal[38]) to overcome this boundary and reduce the data size
 214 of the measured signal. However, these methods presume ideal conditions
 215 for the measurement, a sinusoidal vibration of the structure of a constant
 216 amplitude, which is cumbersome for the evaluation of large structures.

217 The second parameter is the time of measurement, which influences the
 218 resolution of the FFT and, therefore, how precise the sidebands are repre-
 219 sented. Ideally, longer measurement times increase the resolution and hence
 220 the robustness of the data acquisition. However, this is only valid for an
 221 ideally generated modulation. The variant conditions of real applications

222 require the measurements to be timed at recurring conditions. Hence, re-
223 trieving as much information as possible from only short signals is relevant
224 for applying VAM outside the laboratory.

225 If we assume an ideal periodic signal, a traditional VAM has to be mea-
226 sured over a full period T_p of the low-frequency to include all information.
227 For this measurement, the frequency and, more importantly, the amplitude
228 of the low frequency should closely match the prior measurements for com-
229 parable results. However, the synthetic VAM can be generated from only
230 one period T_c of the high-frequency of each stress level n . Since $f_c \gg f_p$,
231 the measurement times can be reduced significantly, resulting in much less
232 data to be saved and transmitted. Furthermore, the measurements can be
233 performed at defined stress states. Hence, the overall amplitude only has to
234 surpass the defined measurement stresses to achieve comparable VAM mea-
235 surements, independent of the actual frequencies present in the structure.

236 Generation of a synthetic signal is explained by using data chunks from
237 the steady-state measurement with a sampling rate above Nyquist's theorem.
238 Since a steady-state of the carrier at each stress measurements is assumed, the
239 carrier of a defined frequency can be recreated from just a few points. Since
240 the frequency is defined by the exciting piezoceramics, only the amplitude
241 and phase of the mean-centered sine must be determined. Hence, a minimum
242 of $p = 2$ measurement points would be sufficient to approximate the steady-
243 state sine and create a synthetic signal with an arbitrary length. [52] By
244 increasing the number of points p , a sine can be fitted to the measurement
245 or more sophisticated methods such as STFT can be applied to calculate
246 these values with a higher tolerance to noise, resulting in higher precision.

247 Consequently, downsampling of the existing measured signals was performed
248 by selecting every 10th or 100th value in order to mimic signals sampled at
249 $f_s = 200$ kHz or even $f_s = 20$ kHz which is less than 0.05 of the dedicated
250 Nyquist rate.

251 Since one of the main goals of this paper is to demonstrate the possibilities
252 a synthetic VAM computation offers over performing dynamic VAM, the
253 shown experimental results stem from $p = 8$ measured points of the high-
254 frequency measurements s_c at the levels σ_l . Furthermore, we evaluate the
255 difference between using the original signal and a signal downsampled by a
256 factor of 100 (i.e. $f_s = 20$ kSa/s).

257 *2.3. Specimen manufacturing and testing*

258 Two types of specimens were produced and tested experimentally to val-
259 idate the proposed synthetic VAM. Two aluminum open-hole tension speci-
260 mens were made in accordance to the work of Dorendorf *et al.*[44] and Oppen-
261 mann *et al.* [38] and similar to the specimen tested by Ramezani. [43] They
262 were cut from a cold-pressed aluminum 7075 plate with a hole of 4.5 mm
263 diameter, drilled in the center. The dimensions are shown in the upper half
264 of Figure 4.

265 Additionally, two specimens were made from a glass fiber reinforced poly-
266 mer laminate (GFRP) in a cross-ply layup that has been manufactured in
267 a resin transfer molding process. Dry UT-E500 E-glass fibers (Gurit) were
268 arranged in a $[0, 90_3]_s$ layup and impregnated with a low-viscosity epoxy sys-
269 tem (RIMH135 and RIMR137 from Hexion). The plate was cured at 50 °C for
270 24 h, and post-cured for 15 h at 80 °C outside the mold in compliance with the
271 datasheet. The fully cured plates were cropped to the specified dimensions

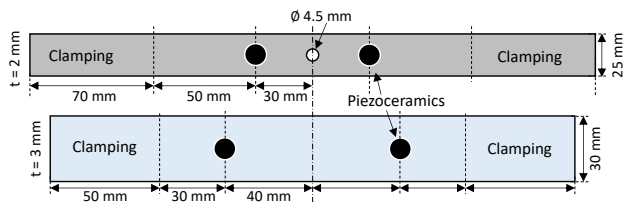


Figure 4: Dimensions and positioning of the piezoceramics for the aluminum (upper) and GFRP $[0, 90_3]_s$ specimens (lower).

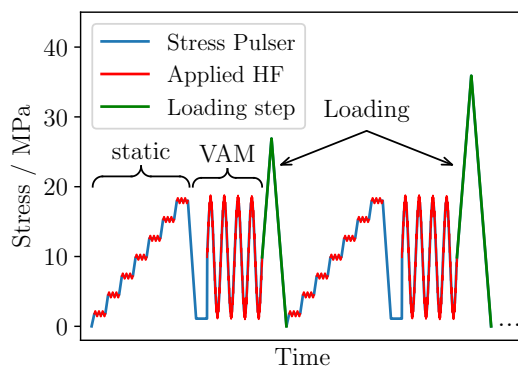


Figure 5: Method of testing to compare the conventional VAM and the synthetically created VAM, combined from the shown static measurements. The blue curve depicts the applied stress with the pulsing machine, and the red curve indicates when the high frequency is excited.

272 with a water-cooled corundum saw blade (ATM Brillant 265). Afterwards,
 273 the cutting surface was polished with 1000-grit sandpaper to reduce crack
 274 initiations from the edges to a minimum. Vibroacoustic measurements and
 275 specimen tests were performed for both materials on a servo-hydraulic test-
 276 ing machine (Instron) with a loading capacity of 40 kN and controlled by the
 277 Instron WaveMatrixTM software. Due to different material properties, failure
 278 mechanisms, and specimen types of the two materials, the testing strategy
 279 was slightly adapted for each material. For the aluminum specimen, a fa-

280 tigue loading was tested in the tension-tension regime with a maximal stress
281 of $\sigma_{\max} = 150$ MPa, a fatigue frequency of $f_{\text{fatigue}} = 5$ Hz and a stress ratio
282 of $R = \frac{\sigma_{\min}}{\sigma_{\max}} = 0.1$ to achieve a lifetime of approximately 30 000 cycles. Prior
283 to the fatigue loading, and after every 2000 cycles, a conventional dynamic
284 VAM ($\sigma_{f_p} = 3 - 30$ MPa) was performed. Additionally, the high-frequency
285 vibration was measured for three static loads ($\sigma_{\text{static}} = 3, 16, 30$ MPa) to
286 compare the synthetic VAM with the conventional method. The process was
287 repeated until the failure occurred. Aluminum specimens were tested over
288 their lifetime to evaluate whether the synthetic VAM can track the damage
289 state and MI of a metallic structure.

290 For the GFRP specimen, increasing loads were applied stepwise to intro-
291 duce defects to the material. When reaching the so-called "knee-stress" of
292 a cross-ply laminate, matrix cracks in between the 90° fibers occur (inter-
293 fiber failure), resulting in a decreased Young's modulus. Hence, after each
294 conventional VAM ($\sigma_{f_p} = 2.2 - 22$ MPa) and the static measurements for
295 the synthetic VAM ($\sigma_{\text{static}} = 2.2, 5.6, 8.9, 12.2, 15.6, 18.9, 22$ MPa), a uniaxial
296 tensile test was performed with a load increase of 5 MPa/s until the defined
297 stress in the specimen was reached. Subsequently, the maximal load was
298 increased until multiple cracks occurred in the specimen, enabling to de-
299 rive a correlation between the number of cracks in the FRP specimen and
300 the conventional and synthetic VAM results. The measurement procedure is
301 illustrated for the GFRP specimens in Figure 5.

302 The high-frequency carrier s_c of all specimens was introduced and mea-
303 sured with piezoceramic actuator disks (PIC255, 10 x 2 mm from PI-Ceramics)
304 which were applied to the specimen by double-sided tape (Tesa 56172). Com-

305 pared to a 2k-adhesive, the vibrations excited in the specimen and, conse-
306 quently, the voltages measured at the receiving piezoceramic are reduced due
307 to the adhesive tape, resulting in a slightly increased signal-to-noise ratio.
308 However, the overall information quality is comparable, while the adhesive
309 tape makes the actuators reusable.

310 To excite the piezoceramic actuators and perform the data acquisition, a
311 NI-USB 6366 data acquisition board (National Instruments) was used with
312 a sampling rate of $f_s = 2 \text{ MSa/s}$. Controlled by Matlab, the generated signal
313 from the NI was twofold amplified (BUF634 with an additional low-pass of
314 420 kHz) to a $12 V_{pp}$ sine. The carrier frequency was determined as the range
315 of the strongest response in a linear frequency sweep between 1–300 kHz.
316 Since the resulting modulation depends on several conditions, as summarized
317 by Lim *et al.* [22], a frequency range of $f_c = 185 - 215 \text{ kHz}$ was chosen. In
318 this range, 61 carrier excitation frequencies f_c have been measured with a
319 spacing of 500 Hz. For all specimens, the dynamic VAM measurements were
320 acquired for a duration of 2 s, and the static measurements for 1 s. Data with
321 sampling rates below $f_s = 2 \text{ MSa/s}$ are generated by selecting the signal at
322 every 10th or 100th value.

323 3. Results and Discussion

324 Before validating the synthetic VAM algorithm with actual measure-
325 ments, a proof of concept was conducted on a signal in which the am-
326 plitude and phase were modulated simultaneously. For both modulations
327 (see eq. A.3 for AM and A.4 for PM in the Appendix), the modulating
328 term $\cos(\omega_p t + \phi_p)$ is quantized into n steps, where n was increased from

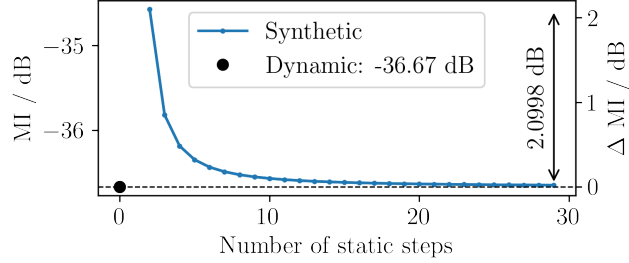


Figure 6: The MI of the synthetic signal converges with an increasing number of steps to the MI of the dynamic measurement.

329 $2 \leq n \leq 30$. For every n , the MI was calculated from the static measure-
 330 ments and compared to the dynamic result. The resulting FFT for $n = 2$ has
 331 already been shown (Figure 3) and was related to the nature of square waves.
 332 As expected, the MI of the synthetic square wave is offset by $20 \cdot \log_{10}(4/\pi) =$
 333 2.098 dB compared to the dynamic VAM. Increasing the number of quantiza-
 334 tion steps results in convergence to the MI of the dynamic signal, as shown in
 335 Figure 6. Also the higher-order sidebands converge towards the values found
 336 in the dynamic measurements and could be used for further evaluations.

337 In the following, our synthetic method is applied to GFRP and aluminum
 338 specimens to compare the resulting modulation to the dynamic VAM that
 339 is traditionally used. The damage was introduced into the GFRP specimens
 340 with subsequent tensile testing at increasing loads, and the resulting cracks
 341 were quantifiable by eyesight. Therefore, both methods can be easily com-
 342 pared and related to the number of cracks. VAM measurements below the
 343 "knee-stress"—which indicates inter-fiber damage of a cross-ply laminate—
 344 were performed to acquire a baseline. When the specimen was loaded with a
 345 tensile stress of 56 MPa, the "knee-stress" was surpassed and first inter-fiber
 346 cracks appeared. The amount of cracks after each loading cycle is shown

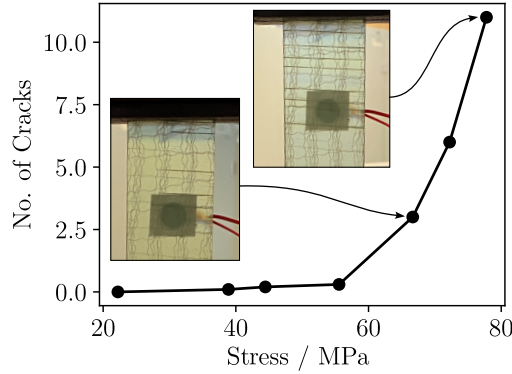


Figure 7: Number of cracks in the 90° layer after each uniaxial loading of the GFRP specimen. The specimen with the manifestation of the cracks is illustrated after two loadings.

347 in Figure 7. In the insets of Figure 7, depicting two different states of the
 348 specimen, the severity and density of inter-fiber cracks is visible by darker
 349 horizontal lines. Note that the first cracks after 56 MPa penetrated the spec-
 350 imen's 90° layers only over half the width but evolved into full cracks at the
 351 next stress level of 69 MPa.

352 Figure 8 (left) compares the measured vibrations with a carrier frequency
 353 of $f_c = 185$ kHz before and after a damage is introduced into the specimen.
 354 Note that the measured signals are almost symmetrical. Therefore, the dif-
 355 ferences between the synthetic signals are shown for $n = 2$ in the upper half
 356 and $n = 3$ in the lower half. To further illustrate the synthetic process, the
 357 signal chunks of the different stress levels are differentiated by color.

358 As expected, the maximal and minimal amplitudes of the dynamic mea-
 359 surements coincide with the amplitudes of the static measurements. When
 360 comparing the amplitudes with respect to the damage and stress, the mea-
 361 sured voltage at a stress of 120 MPa is higher compared to the stress state

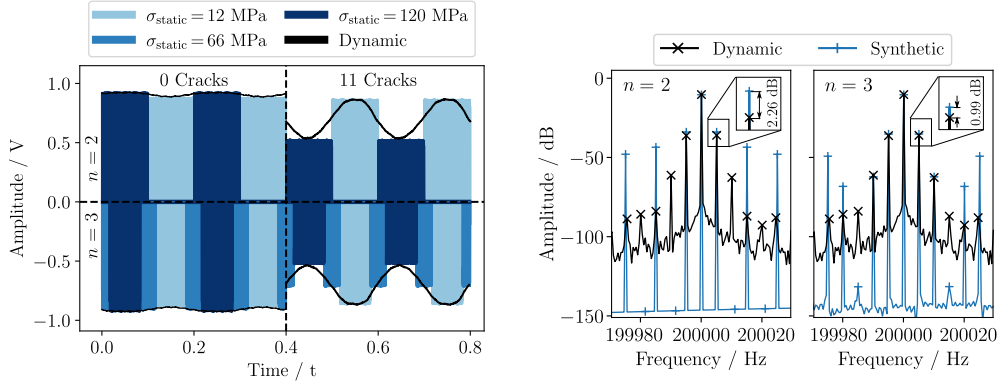


Figure 8: Synthetic generated signals (left) of a GFRP specimen from $n = 2$ (upper) and $n = 3$ (lower) static measurements for the pristine state compared to 11 cracks. The envelope of the dynamic VAM is given as black line. For $n = 2$ and $n = 3$ the frequency spectrum of the synthetic and dynamic VAM is compared (right). The insets show the enlargement of the first sidebands with the mean deviation of this specimen.

362 of 12 MPa for the undamaged specimen. However, this proportion at the
 363 shown frequency reverses after introducing a few cracks. Interestingly, the
 364 measured amplitude at the low stress of 12 MPa stays constant, whereas the
 365 amplitude at the higher stress is significantly reduced due to the cracks. This
 366 observation is in accordance with the often-mentioned bi-linear stiffness be-
 367 havior explaining a VAM by a stress-related crack opening. However, there
 368 are also frequencies ($f_c = 194$ kHz) where the amplitude at the higher stress
 369 state remains constant, while the measured voltage at 12 MPa is decreased.
 370 This duality supports and underlines the physical explanation of VAM given
 371 by Dorendorf *et al.*, [44] which states that the maximal and minimal ampli-
 372 tude of the modulated signal s_r can coincide with either the highest or lowest
 373 stress. The amplitude of the carrier depends on whether the ratio between
 374 carrier frequency and the natural frequency f_c/f_{eigen} is greater than or less

375 than one and, hence, which eigenmode is primarily contributing to the sys-
 376 tem response. Moreover, a phase shift occurs between the alternating and
 377 combined signal fragments, which, as a result, also incorporates angle modu-
 378 lation into the synthetic method. As before with the amplitude modulation,
 379 the change in phase appears stepwise, resulting in shifted values compared
 380 to those of the dynamically modulated phase.

381 Figure 8 (right) illustrates the frequency domain around f_c for both syn-
 382 thetic signals compared to the dynamic measurement. The results are in
 383 accordance with observations made for ideal modulated signals, as shown in
 384 Figure 3 (right). For $n = 2$, the first sidebands are offset from the dynamic
 385 signal by about $\Delta\text{MI} = 2.1$ dB which is reduced to $\Delta\text{MI} = 0.85$ dB for the
 386 three-step signal. Although the higher-order sidebands of the two-step signal
 387 behave according to eq. 2, the three-step synthetic VAM also approximates
 388 the second sidebands with reasonable accuracy by reducing the quantization
 389 error. The disagreement of the third and further sidebands could be improved
 390 by computing the synthetic VAM for more stress levels.

391 The differences between a synthetic and dynamic MI are shown in Fig-
 392 ure 9 for three synthetic signals with $n = 2, 3, 7$ and a range of carrier
 393 frequencies. The synthetic VAM was generated from 8 points of a signal
 394 sampled at $f_s = 2$ MSa/s. As expected, the MI is offset compared to the
 395 dynamic measurements, which mainly results from the synthetic generation
 396 itself. Using only two static measurements $n = 2$ results in a mean off-
 397 set of $\Delta\text{MI}_2 = 2.254 \pm 0.45$ dB, which is close to the expected offset of
 398 $\Delta\text{MI}_{2,\text{ref}} = 2.098$ dB. Therefore, there is a mean error of 0.154 dB, which
 399 can be expressed as 0.73% compared to the mean dynamic MI. This off-

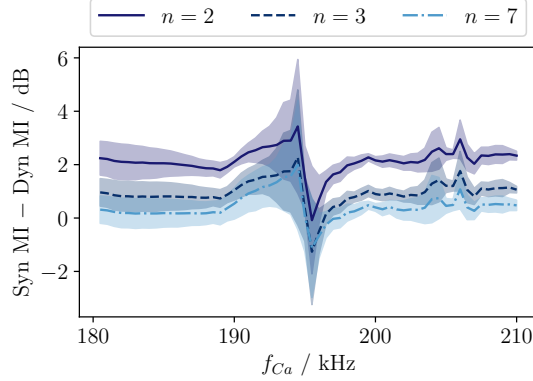


Figure 9: Deviation of the synthetic MI compared to the conventional measurement. Shown are the mean error and its standard deviation of all tensile loadings. Note that the error is the deviation from the expected offset, which for $n = 2$ is $\Delta\text{MI}_{2,\text{ref}} = 2.01$ dB.

400 set can be reduced by using $n = 3$ or even $n = 7$ static stress levels to
 401 $\Delta\text{MI}_7 = 0.44 \pm 0.44$ dB for the synthetic signal generation. Hence, the error
 402 is reduced to 0.12%, which underlines the relevance of this method. More
 403 results are given in Table B.2 *ff.* of the Appendix. Furthermore, Figure 9
 404 indicates, that the mentioned errors result mainly from two frequency ranges
 405 around the resonances, which are at 196 kHz and 205.5 kHz for all n and
 406 result in a minimal modulation. This can be seen in the SI (Figure 1),
 407 where the MI is plotted across the entire frequency spectrum. Interestingly,
 408 the synthetic MI on the left flank of the resonance around 196 kHz is sig-
 409 nificantly increased and on the right side of the resonance decreased. By
 410 choosing a suitable frequency range between 180–190 kHz deviations are re-
 411 duced. It should be mentioned that the areas around the resonances are also
 412 commonly avoided for evaluating a conventional VAM in the literature or
 413 compensated for by averaging over a frequency range, as proposed by several

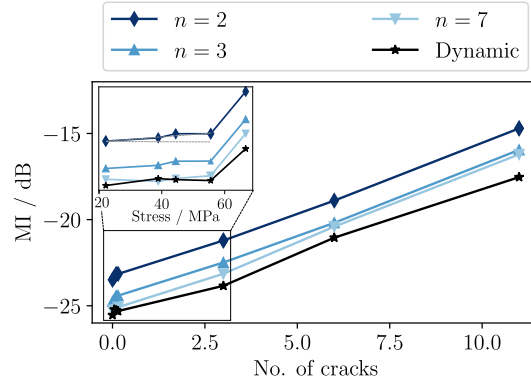


Figure 10: In the GFRP cross-ply laminate, the averaged MI increases linearly with each inter-fiber crack. The mean MI over all measured frequencies for a synthetic signal with various steps (2,3,7) is compared to the dynamic measurement. The inset, shows the same graphs plotted over the max. applied stress, to separate the first four measurements before the first inter-fiber crack appeared, to highlight the increase in MI with each subsequent loading.

414 publications. [7, 23]

415 When averaging the entire frequency range, the MI increases almost lin-
 416 early with every new inter-fiber crack in the specimen, as shown in Figure 10
 417 for the dynamic measurement and synthetic signals. Interestingly, the syn-
 418 thetic MI evolves more linearly. The synthetic MI is able to robustly detect
 419 cracks in GFRP specimen and the difference between the synthetic and dy-
 420 namic MI is negligible compared to the modulation increase per crack. Fur-
 421 thermore, there is an increase in MI even before the first inter-fiber crack was
 422 detected, as enlarged in the inset of figure 10. This increase could indicate
 423 some first microcracks in the composite or damages due to the clamps of
 424 the testing machine. However, the increase in MI due to the first inter-fiber
 425 cracks is significantly higher. All values for the different number of levels

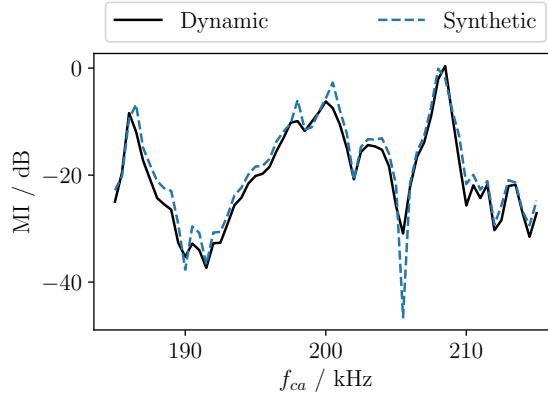


Figure 11: MI of an aluminum specimen (2) after ten thousand fatigue cycles. The dynamic measurement (continuous) is compared to the synthetic computation (dashed).

426 n , the used sampling frequency f_s , and the number of points used p can be
 427 found in the Appendix in Table B.2.

428 The results of the second GFRP specimen are closely comparable to the
 429 presented ones. The deviations are minimally different with an error of 0.97 %
 430 for $n = 2$. Further values can be found in Table B.3.

431 Compared to the tested GFRP specimens, the MI of aluminum speci-
 432 mens fluctuates over 30 dB across the measured carrier frequencies, resulting
 433 from more pronounced resonances in the material. A comparison between a
 434 synthetic ($n = 2$, $f_s = 2 \text{ MSa/s}$, $p = 8$) and dynamic MI after ten thousand
 435 fatigue cycles is shown in Figure 11. Similarly to the GFRP specimen, the
 436 highest deviations occur in the vicinity of the extremes of either weak or
 437 severe modulations (e.g., 206.5 kHz) and result in a slightly increased error
 438 from -1.3 to -1.9 %. However, these errors (around 0.34 dB) are neglectable
 439 compared to the MI increase of 10–30 dB which is shown in literature [20, 38]
 440 as a result of a relevant damage. The mentioned mean deviations and errors

Table 1: Mean deviation and its standard deviation of the tested specimen generated from each $p = 8$ points of $n = 2$ static load levels.

Specimen	Mean deviation	Mean error	
	[dB]	[dB]	[%]
GFRP 1	2.254 ± 0.447	0.154	0.73
GFRP 2	2.302 ± 0.418	0.202	0.97
Aluminum 1	1.828 ± 1.329	-0.272	-1.26
Aluminum 2	1.761 ± 0.804	-0.339	-1.86

441 are summarized in Table 1 and all values can be found in Table B.4 and B.5.

442 The frequencies with a large error, which occur in the vicinity of mini-
 443 mal or maximal modulation, can be explained by inspecting the raw signal.
 444 Here, the assumption of a quasi-linear change in wave propagation due to
 445 the periodically altered stress is no longer valid, resulting in signals with a
 446 non-sinusoidal envelope.

447 3.1. Distorted envelope functions

448 Besides the near-ideal MI concordance between synthetic and dynamic
 449 measurements in most cases, we have found frequencies where the synthetic
 450 method differs from the expected offset. In these cases, the envelope function
 451 of the resulting dynamic signal s_r does not have a nearly sinusoidal form. As
 452 suggested by Dorendorf *et al.*,[44] the steady-state dynamic magnification
 453 factor of a damped oscillation model—which is proportional to the ampli-
 454 tude acquired in our experiments—depends on the excitation frequency in
 455 relation to the natural frequencies (and modes) of the system. Due to the
 456 periodical applied stress in VAM, the natural frequencies of the specimen are

457 periodically shifted, resulting in a periodically changed dynamic magnifica-
458 tion factor. In the vicinity of these natural frequencies, or exactly between
459 two of them, the dynamic magnification factor behaves strongly nonlinear
460 over one period of the low frequency T_p . Hence, the envelope function of
461 the measured response s_r exhibits more than two extreme values during that
462 period T_p .

463 Figure 12 shows a numerical system response of an aluminum specimen
464 which was modeled with the finite element software Abaqus CAE and is
465 demonstrating this phenomenon. The model’s geometry is equivalent to the
466 tested aluminum specimens and the modeling details are described in more
467 detail in the work of Dorendorf *et al.*[44], who has validated the results ex-
468 perimentally using a 3D scanning laser vibrometer (PSV 500-3D-M, Polytec)
469 The model’s parameters are also given in Table C.6 of the Appendix. The
470 specimen’s longitudinal displacement obviously influences the out-of-plane
471 acceleration, which is proportional to the measured voltage, i.e. signal, at a
472 piezoceramic in the prior mentioned experiments. The periodically applied
473 stress results in a severe amplitude modulation of the simulated acceleration.
474 Note that the steady-state amplitudes fit the envelope function at the de-
475 fined stress with negligible error even for frequencies, where the minimum
476 value of the envelope function does not coincide with the minimal or max-
477 imal stress level. As explained in the previous sections—when employing
478 synthetic VAM—it is assumed that the envelope function is near sinusoidal
479 with the minimum and maximum values coinciding with min or max stress
480 or vice versa. Intuitively, when calculating the synthetic VAM ($n = 2$) on
481 these types of signals, the error increases.

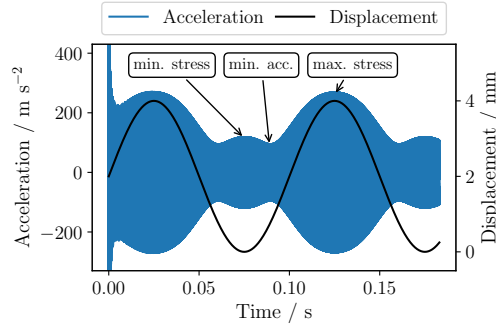


Figure 12: FE simulation of an aluminum specimen with a periodically applied elongation (black line). For certain carrier frequencies ($f_c = 210$ kHz, $f_p = 10$ Hz) the minimal amplitude of the response (blue) is not coinciding with the minimal elongation. (Adapted from [44])

482 In the shown example, the difference is $\Delta MI_2 = 1.68$ dB and further
 483 explains the areas of high errors in the vicinity of the resonance frequencies
 484 (i.e 196 kHz & 205.5 kHz) in Figure 9.

485 Averaging the MI over a frequency range—as proposed in many publications—
 486 minimizes the error due to these frequencies. Furthermore, increasing the
 487 number of levels n reduces this error.

488 3.2. General application for data-reduction

489 Finally, the possibility of further reducing hardware requirements is ad-
 490 dressed, as data reduction is crucial to the deployment of many sensor nodes
 491 to monitor the structural health of large structures with VAM. The most rel-
 492 evant work in this regard is the application of undersampling by Oppermann
 493 *et al.*[38], where a data reduction of 88 – 99.8 % was achieved compared to
 494 a Nyquist-based sampling rate. The lower bound of 88 % was calculated for
 495 the work of Ooijsaar *et al.* [53] who used a pumping frequency of 1455 Hz

496 and a carrier frequency of 50 kHz. Since the greatest common divisor of both
497 frequencies is relatively small, the undersampling method requires a higher
498 sampling rate, resulting in a lower data reduction. For experiments with fre-
499 quencies similar to our work, Oppermann *et al.* could reduce the number of
500 measurement points by 99.8%. With the STFT algorithm applied on a dy-
501 namic undersampled VAM measurement, the MI deviates $\Delta\text{MI} = 0.1668$ dB
502 on average, compared to the traditional evaluation. These remaining 0.2%
503 of the undersampled data still result in 1600 points over a time period of 2 s
504 for which a constant vibration of the structure is assumed. Especially the
505 condition of constant vibration over this time and between the measurements
506 hinders the application of undersampling in actual structures.

507 In contrast, our presented method, the synthetic VAM, is capable of sig-
508 nificantly reducing the number of required points further. For the conducted
509 experiments with pumping frequencies of 5 Hz and the carrier frequency of
510 180 – 215 kHz, only 3 – 8 measured points are needed of at least two stress
511 levels, resulting in a deviation over all frequencies of 0.202 dB for the GFRP
512 and -0.339 dB for aluminum. To evaluate the dynamic VAM, at least 0.2 s
513 sampled at 500 kSa/s are needed, which results in 100 000 data points per
514 measurement and subsequently a reduction of 99.984%.

515 Moreover, with the synthetic method, we can apply the STFT algorithm
516 or a basic upsampling method to the static measurements acquired at a
517 sampling rate much smaller than the excited carrier frequency. This allows
518 for reduced hardware requirements or possible integrations into existing SHM
519 systems. To showcase this opportunity, the original static measurements
520 for the synthetic VAM were downsampled by a factor of 100 resulting in a

521 sampling frequency of $f_s = 20 \text{ kSa/s}$ which is far below the carrier frequency
522 f_c . From eight consecutive points of these downsampled signals, a sine with
523 the original sampling rate is recreated and used for the synthetic process.
524 Since this fitting process is already applied, even when the original sampling
525 rate is used, no further adaptations to the method itself had to be performed.
526 With these eight points and $n = 2$, the error of this method is marginally
527 affected and remains below 0.23 dB for the GFRP specimens and 0.34 dB for
528 the aluminum specimens. So there is still potential to further decrease the
529 required samplingrate.

530 **4. Conclusion**

531 The objective of this work was to explore a new approach to measure
532 the vibroacoustic modulation in large structures, where the measurement
533 conditions are not as ideal and constant compared to laboratory tests with
534 one distinct pump frequency of a defined amplitude. To measure the vi-
535 broacoustic modulation in large structures, the present ambient vibrations
536 are ideally utilized as low-frequency vibration s_p and only the high carrier
537 wave s_c is excited by a piezoceramic. However, it is a cumbersome endeavor,
538 because these vibrations fluctuate in terms of their frequency and amplitude
539 due to the changes in service loads, traffic, and winds. Measured vibrations
540 from e.g. the Köhlbrandbrücke(Hamburg, Germany) demonstrate this and
541 manifest rather in a frequency range (2–20 Hz) than in a single frequency.

542 To overcome these shortcomings of VAM for industrial applications, we
543 propose a new synthetic generation of the signals. It is demonstrated that
544 Lamb wave measurements of less than a millisecond, acquired at several static

545 stress levels, which were interpolated and combined into a synthetic signal
546 are sufficient to robustly approximate the MI and hence detect damages.
547 Moreover, we provide an analysis on how many static stress levels and how
548 many points of the steady state measurement are actually needed in order
549 to achieve an acceptable deviation from the dynamic VAM signal.

550 To synthetically generate more than just the carrier and first sidebands,
551 more than two static stress levels are needed for the synthetic method. Fur-
552 ther validations could be beneficial, since our reference value—the dynam-
553 ically measured MI—is a measured quantity as well and is affected by the
554 machine and measurement settings. However, in the case of the GFRP spec-
555 imen, the synthetically generated MI follows the number of cracks more lin-
556 early than a dynamic MI.

557 With the proposed approach, the data size of each measurement is re-
558 duced significantly. Only 8 points of each level (which is less than 0.016% of
559 data points compared to the conventional methodology) are required. Conse-
560 quently, the energy needed to process the data and the bandwidth required to
561 transmit the data is significantly reduced and further enables the deployment
562 of a self-sufficient sensor network. Hence, the effort and costs of monitoring
563 hardware could be significantly decreased. Furthermore, we have shown that
564 the needed sampling rate for the measurements at each stress level could be
565 reduced by magnitudes below the carrier frequency, while still resulting in
566 sufficient accuracy. Thus, the sensor nodes require lower processing speeds
567 and less energy for data acquisition.

568 This work paves the way and sets the ground for the following material
569 tests and a deployment in larger and more complex structures. We are con-

570 vinced that the individual measurements at several combined static stress
571 states contain the same information as could be extracted from a dynamic
572 VAM measurement. Since the traditional dynamic approach faces the prob-
573 lem of deviations in frequency and amplitude of the pump vibration over
574 time, the synthetic approach is an expedient extension to the VAM method.
575 Hence, robust measurements of the structural health are in reach, regardless
576 of the changing environmental and loading conditions of the structure.

577 *Conflict of interest*

578 The authors declare that they have no known competing financial inter-
579 ests or personal relationships that could have appeared to influence the work
580 reported in this document.

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587 much appreciated.

588 **5. Data Availability**

589 All data is stored on a secure server at the Institute of Polymer and Com-
590 posites, Hamburg University of Technology. The raw data, the python codes
591 for processing and the processed data required to reproduce these findings
592 are available from the corresponding author on reasonable request.

593 **Appendix A. Types of signal modulation**

594 To clarify the synthetic process, and test it with ideally modulated sig-
 595 nals, the different types of signal modulations which can be present in the
 596 VAM signal are presented. A VAM measurement can be the superposition of
 597 amplitude modulation (AM) and angular modulation, which can be further
 598 divided into frequency modulation (FM) and phase modulation (PM). Let
 599 the signal of the ultrasonic carrier wave be defined as

$$s_c(t) = A_c \cdot \sin(\omega_c t) \quad (\text{A.1})$$

600 and the pumping wave introduced by the hydraulic pulsing machine as

$$s_p(t) = A_p \cdot \cos(\omega_p t + \phi_p) \quad (\text{A.2})$$

601 with $\omega = 2\pi f$, the individual frequencies $f_p \ll f_c$ and A as amplitude. Then,
 602 an amplitude-modulated signal can be combined to

$$\begin{aligned} S_{\text{AM}} &= \left[1 + \frac{s_p(t)}{A_c} \right] s_c(t) \\ &= [1 + m_a \cos(\omega_p t + \phi_p)] A_c \sin(\omega_c t). \end{aligned} \quad (\text{A.3})$$

603 The amplitude modulation parameter $m_a = A_p/A_c$ expresses the ratio from
 604 the amplitudes of the pumping wave to the carrier wave.

605 In contrast to the AM, both angular modulations are closely related, as
 606 shown in the following equations. A phase modulation between the given
 607 signals can be expressed as:

$$S_{\text{PM}} = A_c \sin(\omega_c t + m_p \cos(\omega_p t) + \phi_p) \quad (\text{A.4})$$

608 and the frequency modulation, since $s_p(t)$ is a sinusoidal as:

$$S_{\text{FM}} = A_c \sin\left(\omega_c t + \frac{m_f}{\omega_p} \cdot \sin(\omega_p t) + \phi_p\right). \quad (\text{A.5})$$

609 Here, m_p is the phase modulation parameter, and m_f is the frequency modu-
610 lation parameter. When comparing (A.4) and (A.5) it can be concluded that
611 PM is shifted by $\pi/2$ which in addition to multipath propagation through
612 a specimen hinders a separation of FM and PM in an angular modulated
613 signal.

614 **Appendix B. Results for the synthetic recreation of the tested** 615 **specimens**

616 In the following, four tables are presented, in which the MI generated
617 by the synthetic VAM for different numbers of quantization levels n , sam-
618 pling frequencies f_s , and the number of points p is compared to the dynamic
619 measurements. The label of each table is the name of the specimen. Since
620 the specified deviation is expected to have an offset, the corrected deviation
621 is calculated by subtracting the analytically calculated value. The specified
622 error percentage is determined by dividing the corrected deviation by the
623 mean value of the MI in the frequency range.

624

625

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627

628 **Appendix C. Simulation parameter**

629 The simulation was carried out with the finite element software Abaqus
630 CAE. The model's geometry is equivalent to the tested aluminum specimens

Table B.2: Specimen: GFRP 1

n	f_s	p	Mean deviation	Error	
	[kSa/s]		[dB]	[dB]	[%]
2	2000	8	2.254 ± 0.447	0.154	0.73
2	2000	50	2.257 ± 0.448	0.157	0.74
2	20	8	2.256 ± 0.449	0.156	0.73
2	20	50	2.257 ± 0.447	0.157	0.74
3	2000	8	0.988 ± 0.438	0.138	0.61
3	2000	50	0.990 ± 0.437	0.140	0.63
3	20	8	0.989 ± 0.441	0.139	0.62
3	20	50	0.991 ± 0.436	0.141	0.63
7	2000	8	0.437 ± 0.439	0.027	0.12
7	2000	50	0.439 ± 0.437	0.029	0.12
7	20	8	0.438 ± 0.438	0.028	0.12
7	20	50	0.439 ± 0.437	0.029	0.13

631 and the modeling details are described in more detail in the work of Dorendorf
632 *et al.*[44] in Chapter 4. The phenomenon of several modes influencing the
633 signal envelope of the shown signal is already discussed in Fig. 14 of the
634 mentioned publication.

635

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Table B.3: Specimen: GFRP 2

n	f_s	p	Mean deviation	Error	
	[kSa/s]		[dB]	[dB]	[%]
2	2000	8	2.302 ± 0.418	0.202	0.97
2	2000	50	2.237 ± 0.388	0.137	0.66
2	20	8	2.322 ± 0.488	0.222	1.07
2	20	50	2.326 ± 0.421	0.226	1.08
3	2000	8	1.027 ± 0.348	0.177	0.80
3	2000	50	0.964 ± 0.304	0.114	0.52
3	20	8	1.050 ± 0.404	0.200	0.90
3	20	50	1.052 ± 0.336	0.202	0.92
7	2000	8	0.494 ± 0.314	0.084	0.37
7	2000	50	0.413 ± 0.288	0.003	0.01
7	20	8	0.507 ± 0.365	0.097	0.43
7	20	50	0.500 ± 0.302	0.090	0.40

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Table B.4: Specimen: Aluminum 1

n	f_s	p	Mean deviation	Error	
	[kSa/s]		[dB]	[dB]	[%]
2	2000	8	1.828 ± 1.329	-0.272	-1.26
2	2000	50	1.909 ± 1.142	-0.191	-0.89
2	20	8	2.154 ± 1.308	0.054	0.25
2	20	50	1.960 ± 1.088	-0.140	-0.65
3	2000	8	0.630 ± 1.365	-0.220	-0.96
3	2000	50	0.727 ± 1.154	-0.123	-0.54
3	20	8	0.955 ± 1.303	0.105	0.47
3	20	50	0.770 ± 1.101	-0.080	-0.35

Table B.5: Specimen: Aluminum 2

n	f_s	p	Mean deviation	Error	
	[kSa/s]		[dB]	[dB]	[%]
2	2000	8	1.761 ± 0.804	-0.339	-1.86
2	2000	50	1.740 ± 0.805	-0.360	-1.98
2	20	8	1.762 ± 0.795	-0.338	-1.85
2	20	50	1.752 ± 0.775	-0.348	-1.91
3	2000	8	0.492 ± 0.730	-0.358	-1.84
3	2000	50	0.459 ± 0.708	-0.391	-2.00
3	20	8	0.487 ± 0.716	-0.363	-1.86
3	20	50	0.462 ± 0.683	-0.388	-1.99

Table C.6: Material properties of finite elements simulation with Abaqus CAE.

Variable	Variable	Value
Carrier frequency	f_c	210 000 Hz
Pump frequency	f_p	10 Hz
damping ratio	ξ	0.01
Rayleigh damping coefficients	α	80
	β	1.38e-8
Young's modulus	E	$7 \cdot 10^4$ N/mm ²
Poisson's ratio	ν	0.296
density	ρ	2752 kg/m ³

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